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العلوم الأساسية والتطبيقية Basic and Applied Sciences



# Tuning of Morphological, Structural and **Optical Properties of a-Fe<sub>2</sub>O<sub>3</sub>** Nanoparticles by Precursor Concentration

التحكم بالخواص الضوئية والتركيبية للحبيبات الناندية لأكسيد الحديد الهمياتيت باستخدام

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### ABSTRACT

The current study focuses on a dual aim of developing an easy method to synthesise monodisperse hematite iron oxide ( $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>) nanoparticles and controlling their optical, structural and morphological properties. By using a low-cost and eco-friendly room temperature sonication method, a series of iron oxide nanoparticles were prepared on a nano scale, with fine sizes ranging from 4–10 nm. Optical, structural and morphological properties were precisely modified by varying the concentration of the solution of iron chloride (0.01 M, 0.02 M and 0.05 M). X-ray diffraction and Raman spectra revealed that the prepared nanoparticles had a high crystallinity, confirming that they were hematite. The nanoparticles' crystallite sizes increased from 6 to 10 nm with solution molarity. The optical measurements showed that the band gap energy of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanoparticles (annealed) decreased from 2.67 eV to 2.46 eV with the increasing of solution molarity from 0.01 M to 0.05 M, respectively. Results of field emission scanning electron microscopy demonstrated that the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanoparticles appeared as spherical dots, and their sizes increased from 6 to 10 nm with the concentration of the solution. The presented green synthesis might be useful in controlling the functional properties of magnetic nanostructures for a variety of applications.

# 1. Introduction

Hematite ( $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>) considers one of n-type semiconductors with band gap energy around 2.1 eV, which becomes one of the good stable phases thermodynamically of Fe<sub>2</sub>O<sub>3</sub> family. lt is an inexpensive magnetic material and has environmentally benign properties. Hematite has been widely used in numerous applications including magnetic recording media, gas sensors catalysts, lithium ion batteries, biomedicine and bio-diagnostics due to its nontoxic nature, cost effectiveness, and high corrosion resistance (Chen et al., 2010; Cherian et al., 2012; Horak et al., 2017; Khabiri et al., 2020; Khatami et al., 2019; Patil et al., 2018; Ramalingam et al., 2020; Wang et al., 2012; Widayat et al., 2019; Xia et al., 2018; Zhu et al., 2010). Generally, the (nano/micro) materials properties are strongly linked to their morphology and size (Xia et al., 2009). For the utilization of various technological applications, the adjustment of their sizes, shapes and porosity was intensively followed.

Numerous methods have been used for preparing iron oxides nanoparticles (Ibrahim et al., 2018; Lubis et al., 2018; Mufti et al., 2014; Qiu et al., 2011; Rasheed et al., 2016). In the same trend, chemical precipitation, (Sivakumar et al., 2014) solid-state reaction, (Li et al., 2014) hydrothermal method, (Xu et al., 2015) sol-gel method, (Kopanja et al., 2016) and pyrolytic process (Chen et al., 2010) have been used for developing iron oxide nanostructures. However, these synthesis routes appear to be inopportune because of the requirement of complete template removal, high cost and long synthesis cycle in addition to low yield and large amount of solvents as shown in the sol-gel method and micro-emulsion process (Tao et al., 2020). Also, the elimination of the surfactants or shape-control ions is not favorable for the morphology of the final products

اللخص

في الدراسة الحالية، تم تحضير حبيبات متناهية في الصغر في حجم 4-10 نانومتر باستخدام طريقة رخيصة الثمن وقليلة التكلفة وصديقة للبيئة عن طريق الموجات الصُّوتية، وباستخدام تغيير تركيز نترات الحديد، تم التحكم في الخواص الضوئية والتركيبية. نُتائج حيود أشعة إكس وضحتُ وأكدت أنَّ المنتج عالى البلورة ومكون من حبيبات في حجم النانو من نوع الهماتيت. حجم الحبيبات كان يتراوح بين 4-8 نانومتر، وبالمعالجة الحرارية تم زيادة حجم البلورات لتكون في مدى 6-10 نانومتر. القياسات الضوئية أظهرت أن طاقة الفجوة للحبيبات النانوبة 2.67 إلكترون فولت. وبالتحكم في تركيز نترات الحديد، تم تقليل طاقة الفجوة إلى 2.46 إلكترون فولت. نتائج الميكروسكوب الإلكتروني الماسح أظهرت أن الجبيبات النانوية لها شكل دائري وأكدت أن حجم الحبيبات النانوية يمكن التحكم بها بين 6 نانومتر إلى 10 نانومتر اعتمادًا على التركيزات. في النهاية، يمكن أستخلاص أن طرق التحضير الخضراء تكون مفيدة في التحكم في الخواص الوظيفية للتركيبات النانوبة المغناطيسية وبالتالي تكون متاحة لتطبيقات صناعية واسعة.

(Bondarchuk et al., 2017; Zhou et al., 2020). Moreover, the previously mentioned synthetic routes required longer reaction time, high temperature and pressure, high cost and poor safety performance in addition to complicated steps as seen in the hydrothermal techniques and solid-state reactions. In the chemical precipitation process, there are serious agglomerations of particles leading to application limit and wide particle size distribution (Patil et al., 2018; Ren et al., 2017; Ryan et al., 2017).

Therefore, it is essential to develop a simple and adaptable method for preparing  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanostructures with tunable sizes. As compared with other synthesis methods, due to its simplicity, and large-scale production feasibility, the solution-based synthesis technique is more attractive. For the growth of  $\alpha$ - Fe<sub>2</sub>O<sub>3</sub> nanostructures, ultra-sonication synthesis has been shown to be an adaptable method because of its low processing temperature, mass production, high quality, cost effective and environmental benign nature.

In this trend, the current study has used a simple sonication synthesis method for synthesizing  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanoparticles. Their structural, optical, and morphological properties can be easily tuned by the solution concentration. X-ray diffraction, field emission scanning electron microscopy and Raman spectroscopy have used to characterize the prepared nanomaterials. The optical properties were measured and used for calculating the band gap energy for the prepared nanostructures.

# 2. Materials and Methods

A simple sonication method was used for synthesizing  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanoparticles. Analytically pure reagents (A. R.) of FeCl<sub>3</sub>·6H<sub>2</sub>O (purity $\geq$  98%) has been dissolved in de-ionized water (DIW) for forming an aqueous solution at various concentrations: 0.01, 0.02 and 0.05 M of iron chloride. The reaction solution in a beaker was then transferred to an ultrasonication system at room temperature. The reaction has been accomplished for 30 min by adding ammonia (approximately 30%) to the solution at pH=4. After the reaction was completed,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> precursors were precipitated, which was then washed thoroughly few times with ethanol and water. The product was filtrated, dried and grounded. The dried product calcined at 400 °C for 2 hours in a muffle furnace.

X-ray diffraction (XRD) was used to characterize the prepared materials by using X-ray diffractometer, Philips X'PERT MPD PW3040 equipped with Cu K $\alpha$  radiation at  $\lambda$ =1.5406 Å, operating at a current value of 30 mA and a voltage value of 40 kV. The XRD patterns were recorded in the scan ranging from 30° to 65° of 2 $\theta$  under a step size of 0.02° with scan speed of 20/min. Images of field emission scanning electron microscopy (FE-SEM) were collected using a JEOL microscope. A (UV– Vis) spectrophotometer (Shimadzu) was used to measure the room temperature optical absorption spectrum ranging from 200–800 nm. Phonon vibrational measurements of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanostructures were carried out by using micro-Raman spectrometer (Renishaw) with an excitation source of 532-nm of (SSL) solid state primary laser.

# 3. Results and Discussion

The XRD patterns of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanoparticles without calcinations were illustrated in Figure 1. It can be seen that the samples of low concentrations (0.01 M and 0.02 M) consist of large amorphous phases. Whereas the characteristic peaks of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> were observed in the high concentration. The clear broad peaks indicated that minute crystalline  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> is clearly formed in the solution process at RT.



Figure 2. XRD patterns of  $\alpha$ -Fe2O3 nanoparticles calcined at 400 oC for 2 h.



Calculated values of average crystallite size of as-prepared  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanoparticles estimated according to the Scherer equation was found to be ~4 nm, ~5 nm, ~7 nm. After being calcined at 400 °C for 2 h, there is an evident crystallization as shown in the XRD patterns of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanoparticles in Figure 2. The process of crystallization are

well-established at 400 °C with enhancing intensities of some primary peaks and appearing new diffraction peaks. Within the XRD detection limit, it was clear that all the diffraction peaks obtained for all these samples are well matched to the diffraction pattern of the standard JCPDS card No. 89-8104 of pure rhombohedral symmetry of hematite Fe<sub>2</sub>O<sub>3</sub> (Ahmed *et al.*, 2014). No impurity peaks were observed in the diagrams. The sharp and strong diffraction peaks illustrated a high degree of crystallization of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanostructures. Collected values of average crystallite size of the calcined nanoparticles obtained using the Scherer equation was found to be ~6 nm, ~8 nm, ~10 nm for 0.01, 0.02, and 0.05 M concentration, respectively. These results indicated that increasing concentration of iron chloride lead to improve the crystallization of as-synthesized products.

Figure 3. FESEM images of without calcined  $\alpha$ -Fe2O3 nanoparticles with various concentration of (a) 0.01 M (b) 0.02 M, (c) 0.05 M, and (d) table showing the variation of grain size with concentration.



Figures 3 (a-c) demonstrates the FE-SEM images of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> with 0.01, 0.02, and 0.05 M concentration without calcined (as prepared). It is obvious from Figure 3(a) that the agglomerated fine particles of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> have an average diameter of ~4 nm, while for 0.02 M, the average diameter of ~5 nm (see Figure 3(b)), and for 0.05 M, diameter of ~7 nm (see Figure 3(c)), respectively. These images indicate that crystallization was not completely occurred, which resulted to agglomerate particles with smaller size. Interestingly, after the calcinations, evident change in morphology was observed as shown in Figure 4.

Figure 4. FESEM images of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanoparticles calcined at 400 °C for 2 h with various concentration of (a) 0.01 M (b) 0.02 M, (c) 0.05 M, and (d) table showing the variation of grain size with concentration.



Figures 4(a-c) show FESEM images of the calcined samples at 400 °C for 2 h of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> with a concentration of 0.01, 0.02, and 0.05 M, respectively. Well dispersed spherical nanoparticles with diameter of ~6 nm were obtained for  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> with 0.01 M (see Figure 4(a)). By increasing solution concentration, size of the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> particles was

found to increase to ~8 nm for 0.02 M (see Figure 4(b)), ~10 nm for 0.05 M (see Figure 4(c)), respectively. The different solution concentration of iron precursor resulted to increase size of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> while keeping the experimental conditions fixed for all the samples.

Figure 5. (a) UV-Vis absorption spectra of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanoparticles calcined at 400 °C for 2 h, (b) corresponding energy bandgap plot, (c) table showing the variation of bandgap with concentration, and (d) Plot of energy bandgap and crystallite size as a function of



Optical properties of the prepared samples have been studied by UVvisible absorption spectra as shown in Figure 5 (a). It has been obviously obtained from the figure that the absorption peaks demonstrated a red shift as the size of the nanoparticles increased with the increase of concentration. The energy band-gap of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanoparticles was obtained by using this equation (Parida *et al.*, 2011).

$$\alpha h v = \alpha (h v - E_g)^n \tag{1}$$

where  $\alpha$  is a well know coefficient of absorption, A is constant,  $\nu$  is the light frequency,  $E_g$  is defined as the energy band-gap, h is Planck's constant and n is the transition kind of the semiconductor. The optical band gap was found by plotting  $(\alpha h \nu)^2$  vs. hV. Figure 5(b) represents the plot of band-gap of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanoparticles with various concentrations. The energy bandgap of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> with 0.02 M and 0.05 M concentration are 2.67 and 2.46 eV, respectively, showing a clear red shift as the sizes increased. Therefore, a comparison to the bulk obtained value  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (2.2 eV) (Zhu *et al.*, 2012), shows that the absorption band edge optically for the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> shows red shift with reference to that of the bulk  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>. This shift might be ascribed to the clear effect of quantum size (Brus, 1996) which resulted from the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanoparticles. Figure 5(d) shows that the bandgap value decreased and the size increased with the increase of solution concentration.

Figure 6. Room temperature Raman spectra of $\alpha$ -Fe <sub>2</sub> O <sub>3</sub> nanoparticles calcined at 400	)°C
for 2 h with various concentration.	



To understand the quality and confirm crystal phases of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanoparticles, structure sensitive Raman spectroscopic measurements have been performed. Figure 6 showed the Raman spectra of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanoparticles which recorded at room temperature. The most familiar iron oxide on earth is  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanoparticles which crystallized in a corundum-type structure. In the standard Raman spectrum of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, the expected phonon lines are seven in which two A1<sub>a</sub> modes and five  $E_{\sigma}$  modes are well known (Chen *et al.*, 2010). It is clear from figure 6 that the peaks that appear in the spectra of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanoparticles with 0.01 M concentration at 212, 271, 386, and 580 cm<sup>-1</sup> connected with the specific peaks of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, i.e. the peak positioned at 212 cm<sup>-1</sup> relate to the A1 $_{\sigma}$  mode. The three peaks at about 271, 386 and 580 cm<sup>-1</sup> are attributed to the Eg mode (Chen et al., 2010). Nevertheless, with the increase in concentration, some shifts towards higher wavenumbers were evidenced which might be due to increase in size of the nanoparticles. These findings concluded that a highly crystalline  $\alpha$ - $Fe_2O_3$  can be easily produced by one-step sonication method.

### 4. Conclusion

In summary, an easy, cost-effective and environmentally benign sonication process has been used to prepare  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanoparticles with tunable size in guantum confinement range. FESEM images showed that the prepared samples were in quantum regime, and the size of the nanoparticles increased from 4 nm to 11 nm with the increase value in concentration from 0.01 M to 0.05 M. XRD and Raman results revealed that  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanoparticles possessed crystalline nature with rhombohedral symmetry. The optical appearance properties demonstrated that the energy band-gap of the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanoparticles decreased from 2.67 to 2.46 eV associated with the increase value in concentration resulted to an obvious red shift. Finally, the current study concluded that the possibility to tune the particles size, energy band gap, and the crystallinity of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanoparticles have been achieved by using a simple and cost effective sonication method. This method might be useful for preparing magnetic and non-magnetic nanostructures with tunable properties for optical applications.

### Bios

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